

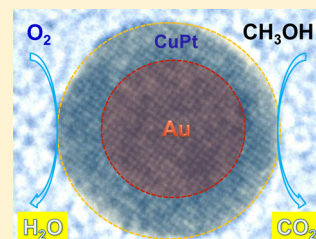
1 Core/Shell Au/CuPt Nanoparticles and Their Dual Electrocatalysis for 2 Both Reduction and Oxidation Reactions

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7 **ABSTRACT:** We report a facile synthesis of monodisperse core/shell 5/1.5 nm Au/CuPt
8 nanoparticles by coreduction of platinum acetylacetonate and copper acetylacetonate in the
9 presence of 5 nm Au nanoparticles. The CuPt alloy effect and core/shell interactions make
10 these Au/CuPt nanoparticles a promising catalyst for both oxygen reduction reactions and
11 methanol oxidation reactions in 0.1 M HClO₄ solution. Their specific (mass) reduction and
12 oxidation activities reach 2.72 mA/cm² (1500 mA/mg_{Pt}) at 0.9 V and 0.755 mA/cm² (441 mA/
13 mg_{Pt}) at 0.8 V (vs reversible hydrogen electrode), respectively. Our studies show that the
14 existence of the Au nanoparticle core not only minimizes the Pt usage but also improves the
15 stability of the Au/CuPt catalyst for fuel cell reactions. The results suggest that the core/shell
16 design is indeed effective for optimizing nanoparticle catalysis. The same concept may be extended to other multimetallic
17 nanoparticle systems, making it possible to tune nanoparticle catalysis for many different chemical reactions.



18 ■ INTRODUCTION

19 Coupling fuel oxidation and oxygen reduction reactions at or
20 near ambient conditions is an important approach to convert
21 chemical energy stored in fuel to electric energy.¹ To ensure
22 easy electron flow from fuel (via fuel oxidation) to oxygen (via
23 oxygen reduction) at lower oxidation and reduction over-
24 potentials, an efficient catalyst must be present to catalyze each
25 of the two reactions.² Traditionally, nanoparticles (NPs) of Pt
26 and its alloys with Ru are selected as such catalysts due to their
27 relatively high activity and durability for both reactions in acidic
28 media. Recently, the need to minimize the use of Pt has
29 motivated the extended search for more efficient Pt NP
30 catalysts with parameters tuned specifically for NP shapes,³ Pt
31 alloying with other early transition metals,⁴ and core/shell
32 structures.⁵ Among these three classes of catalysts studied,
33 shape and alloy composition-controlled catalysts often show
34 some impressive enhancement in initial activity but have very
35 limited improvement in stability unless the catalyst is prepared
36 to have one-dimensional (1D) nanostructure.⁶ Core/shell NPs,
37 on the other hand, are found to be more promising as a robust
38 catalyst. These Pt-based core/shell NPs with a non-Pt core not
39 only maximize Pt exposure to reactants but also enhance Pt
40 catalysis via strong core/shell interactions.⁷ However, because
41 of the difference in the chemical reaction nature between
42 oxygen reduction and fuel oxidation reactions, these core/shell
43 catalysts are often applied to a specific one. Therefore, different
44 kinds of Pt catalysts have to be developed to improve the
45 efficiency of these two reactions.

46 Herein we report a controlled synthesis of core/shell Au/
47 CuPt NPs and demonstrate their enhanced dual catalysis for
48 both oxygen reduction reaction (ORR) and methanol oxidation
49 reaction (MOR) in 0.1 M HClO₄ solution. In studying core/
50 shell NPs as a robust catalyst for fuel cell reactions, we noticed

that Au/FePt₃ NPs with a 1.5 nm FePt₃ shell showed much
51 improved activity and durability in catalyzing ORR in 0.1 M
52 HClO₄ solution^{5a} while FePt/PtAu with FePt in face-centered
53 tetragonal structure and Au segregating around the Pt shell
54 were highly efficient for formic acid oxidation reaction (FAOR)
55 and MOR.⁸ However, they became ineffective if their role was
56 switched. Interestingly, core/shell CuPt/Pt NPs made from
57 controlled dealloying of Cu from CuPt NPs showed enhanced
58 ORR catalysis,^{7a} and cubic CuPt NPs were found to be active
59 for FAOR and MOR with enhanced CO tolerance.⁹ This
60 indicated that CuPt as a catalyst component was capable of
61 serving as a catalyst for both ORR and fuel oxidation reactions.
62 Therefore, we synthesized core/shell Au/CuPt NPs via a seed-
63 mediated growth method and expected that the CuPt shell
64 would exhibit dual catalysis and the Au core would further
65 enhance the catalysis by stabilizing the CuPt shell. Our tests
66 showed that these Au/CuPt NPs were indeed efficient in its
67 dual catalysis for both ORR and MOR. Its ORR and MOR
68 specific (mass) activities reached 2.72 mA/cm² (1500 mA/
69 mg_{Pt}) at 0.9 V and 0.755 mA/cm² (441 mA/mg_{Pt}) at 0.8 V (vs
70 reversible hydrogen electrode, RHE), respectively. 71

■ EXPERIMENTAL SECTION

72 **Materials.** 1-Octadecene (ODE), borane *tert*-butylamine complex
73 (BBA), oleic acid (OA), 1,2,3,4-tetrahydronaphthalene (tetralin),
74 platinum acetylacetonate (Pt(acac)₂), copper acetylacetonate (Cu-
75 (acac)₂), 1,2-hexadecanediol (HDD), Nafion 117, and carbon black
76 (Ketjen EC 300J) were all from Sigma Aldrich. Oleylamine (OAm)
77 was from Acros Organics. H₂AuCl₄·3H₂O was from Strem Chemicals.
78 All chemicals were used as received. 79

Received: January 19, 2014

Instruments. Transmission electron microscopy (TEM) images were acquired on a Philips EM 420 (120 kV). High resolution TEM (HRTEM) images were obtained on a JEOL 2010 TEM (200 kV). High resolution high angle annular dark-field scanning TEM (HAADF-STEM) and elemental mapping images were obtained on a JEOL 2200FS microscope with a beam size of ~ 0.8 Å for imaging and ~ 2 Å for chemical analysis. All TEM samples were prepared by depositing a drop of diluted NP dispersion in hexane on a copper grid coated with amorphous carbon. X-ray diffraction (XRD) patterns were collected on a Bruker AXS D8-Advanced diffractometer with Cu K α radiation ($\lambda = 1.5418$ Å). The inductively coupled plasma mass spectrometry (ICP-AES) analyses were performed on a JY2000 Ultrace ICP atomic emission spectrometer equipped with a JY AS 421 autosampler and 2400 g/mm holographic grating. Energy dispersive X-ray spectroscopy (EDS) was carried out on a JEOL JSM-6060 scanning electron microscope (SEM). Samples for EDS were deposited on a graphitized porous carbon support. UV-vis spectra were recorded on a Perkin-Elmer Lambda 35 spectrometer. Electrochemical measurements were carried out on a Pine electrochemical analyzer, model AFCBP1. Ag/AgCl (filled with 0.1 M KNO₃) and Pt wire were used as reference and counter electrodes, respectively. All potentials were calibrated against reversible hydrogen electrode (RHE).

Synthesis of Au NPs. Au NPs were synthesized through adopting the previous protocol.¹⁰ HAuCl₄·3H₂O (0.2 g) was dissolved in tetralin (10 mL) and OAm (10 mL), and the solution was cooled in an ice bath (0 °C). BBA (1 mmol), tetralin (1 mL), and OAm (1 mL) were mixed by sonication and quickly injected into the above solution. The reaction mixture was further stirred at room temperature for 3 h. Au NPs were precipitated by adding acetone and collected by centrifugation. The product was redispersed in hexane and separated by adding ethanol and centrifugation. The final product was dispersed in hexane.

Synthesis of Au/CuPt Core/Shell NPs. Cu(acac)₂ (0.25 mmol) Pt(acac)₂ (0.25 mmol) and HDD (2 mmol) were dissolved in 10 mL of ODE before Au NPs (35 mg) in hexane (1 mL) were added. The mixture was heated to 120 °C under a gentle N₂ flow to remove hexane before it was heated to 200 °C at a heating rate of 2 °C/min and kept at this temperature for 30 min. The reaction mixture was cooled to room temperature. The product was precipitated by 2-propanol and collected by centrifugation. The product was redispersed in hexane and separated by adding ethanol and centrifugation. The final product was dispersed in hexane.

Electrochemical Measurements. Ten milligrams of the as-synthesized NPs and 20 mg of Ketjen-300J carbon support were mixed in 10 mL of hexane and sonicated with a Fischer Scientific FS 110 for 60 min. The product was then suspended in 20 mL of acetic acid at 60 °C overnight under a mild N₂ flow. The catalyst (C-NPs) was separated by centrifugation and washed with ethanol three times before it was suspended in deionized water at a concentration of 2 mg/130 mL. Nafion (0.5% v/v) was added, and the suspension was sonicated for 1 h.

Ten microliters of C-NPs in water was dropped on a rotation disk electrode (RDE) with a glassy carbon surface (5 nm in diameter from Hokuto Denko Corp., Japan). Water was slowly evaporated in the air, and another 10 μ L of C-NPs was dropped on the electrode and dried to ensure a complete coverage of electrode surface.

Surface cleaning was carried out by CV scanning in N₂ saturated 0.1 M HClO₄ at room temperature between 0.05 and 1.20 V at a scan rate of 50 mV/s for 100 scans. CO stripping voltammetry was performed by scanning between 0.05 and 1.10 V in Ar-purged 0.1 M HClO₄ at 50 mV/s after a CO_{adlayer} was formed on the NP surface in the CO-saturated 0.1 M HClO₄ at 0.10 V. Oxygen reduction reaction (ORR) catalyzed by C-NPs was measured by a rotation disk electrode at 1600 rpm and 20 mV/s. The catalyst stability was checked by scanning between 0.60 and 1.10 V at 100 mV/s for 5000 sweeps. Methanol oxidation reaction (MOR) catalyzed by the same C-NPs was evaluated in N₂-saturated 0.1 M HClO₄ containing 0.1 M methanol at room temperature. The CV scan was carried out between 0.3 and 1.2 V with

a scan rate of 20 mV/s. The stability was tested by chronoamperometry at 0.8 V for 1 h.

RESULTS AND DISCUSSION

The core/shell Au/CuPt NPs were prepared by simultaneous reduction of Pt(acac)₂ and Cu(acac)₂ in the mixture of 5 nm seeding Au NPs, HDD, OAm, OA, and ODE at 200 °C. In the synthesis, Au NPs were premade and coated with OAm. HDD served as a reducing agent and OA/OAm as surfactants. This seed-mediated growth condition facilitated CuPt deposition around each Au NP, forming core/shell Au/CuPt with a uniform shell of CuPt. CuPt compositions were controlled by the molar ratio of Cu(acac)₂ to Pt(acac)₂ and analyzed by ICP-MS and EDS. For example, Pt₆₅Cu₃₅, Pt₅₀Cu₅₀, and Pt₃₆Cu₆₄ were obtained from 3:2, 1:1, and 3:5 Pt/Cu ratios, respectively.

Figure 1a,b shows the typical TEM images of the seeding 5 nm Au NPs and the core/shell NPs. After CuPt coating, the

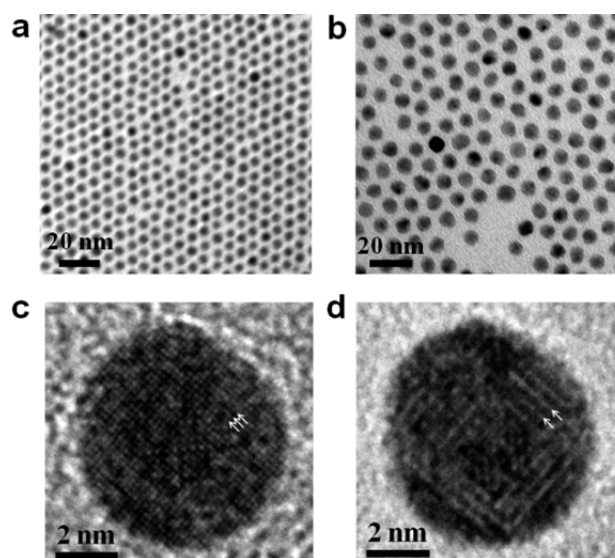


Figure 1. TEM images of (a) 5 nm Au NPs, (b) 7.5 nm Au/Cu₃₅Pt₆₅ NPs. (c, d) HRTEM images of a single Au/Cu₃₅Pt₆₅ NP acquired at a different focus condition.

average size of the core/shell NPs was increased from 5 ± 0.5 nm (Au NPs) to 7.5 ± 0.5 nm (Au/CuPt), indicating the coating thickness was at 1.3 nm. High resolution TEM (HRTEM) was used to analyze the detailed structure of a single NP, as shown in Figure 1c,d. The interface between the Au core and Pt shell cannot be clearly seen because of the close scattering power and lattice parameter of Au and Pt in the core and shell structure. There is also no clear evidence of the existence of misfit dislocation, as observed in the Au/FePt system.¹¹ However, the HRTEM image of the same NP (Figure 1c) recorded at different focus conditions (Figure 1d) shows a long-range ordered structure as highlighted by the arrowheads, which must come from the CuPt alloy structure. The formation of the ordered structure indicates that CuPt is successfully coated over the Au core. The core/shell structure was further confirmed by the linear EDS scan across a single particle (Figure 2). Operated at 300 kV, EDS revealed that the Au peak was 1 nm narrower than the Pt peak at the beginning of the scan (the left side of the line scan, Figure 2a,b), indicating that the coating thickness was around 1 nm. However, at the end of the scan, the NP showed an alloy-type structure (Figure 2b). We also obtained the image of the NP we scanned and noticed

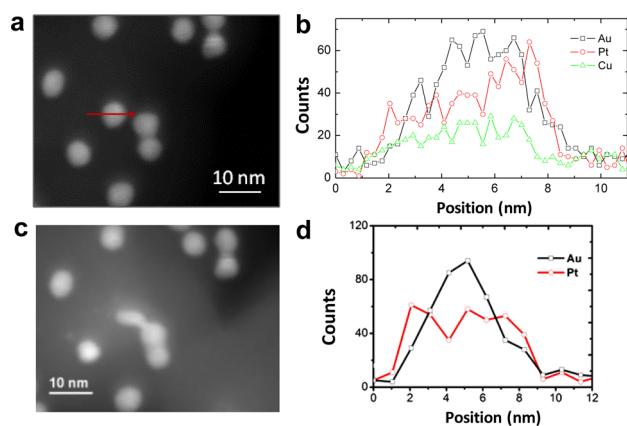


Figure 2. (a) HAADF-STEM image of the Au/CuPt NPs before the EDS scan at 300 kV, (b) line-scan EDS analysis across a single Au/CuPt NP indicated in panel a, (c) HAADF-STEM image of the same Au/CuPt NPs shown in panel a after the EDS scan at 300 kV, indicating the diffusion between the core and shell, and (d) line-scan EDS analysis across a single Au/CuPt NP at 200 kV.

187 the morphology change upon electron beam irradiation (Figure
188 2c). However, when the operation voltage was decreased to 200
189 kV, the EDS analysis showed clearly the core/shell structure
190 (Figure 2d). This indicates that the core/shell Au/CuPt is
191 indeed formed and that the high voltage electron beam used for
192 EDS analysis can trigger the core/shell diffusion into the solid
193 solution structure.

194 In the synthesis, the ratio of Au seeds over Pt(acac)₂ was
195 kept the same and the amount of Cu(acac)₂ was used to control
196 Cu/Pt compositions. Figure 3a is the plot of atomic % of Cu in

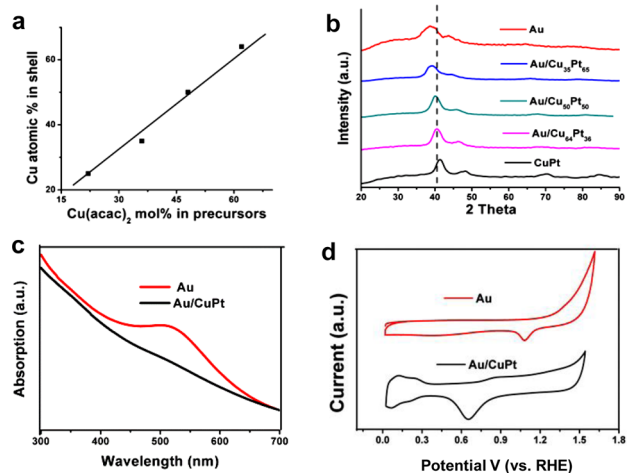


Figure 3. (a) Correlation between the amount of Cu(acac)₂ added and the amount of Cu obtained in the final CuPt shell (b) XRD pattern of Au/Cu_xPt_{100-x} ($x = 35, 50, 64$) core/shell NPs. (c) UV-vis spectra of the 5 nm Au NPs and Au/CuPt NPs in hexane. (d) CVs of the C-Au and C-Au/CuPt in N₂-saturated 0.1 M HClO₄ solution.

197 the CuPt shell vs mol % Cu(acac)₂ used to react with 0.25
198 mmol of Pt(acac)₂ in coating CuPt over Au NPs. The molar
199 percentage of Cu in the shell structure (y -axis) is nearly the
200 same as that of Cu(acac)₂ in the two precursors (Cu(acac)₂ and
201 Pt(acac)₂) (x -axis), which indicates that the Cu/Pt composi-
202 tions are carried over to the final CuPt shell under our synthetic
203 conditions. Figure 3b shows the XRD patterns of the Au/
204 Cu_xPt_{100-x} ($x = 35, 50, 64$) core/shell NPs and pure CuPt.

The (111) diffraction of the Au NPs appears at 38° while that 205
of the Cu₅₀Pt₅₀ NPs at 41.4°. Once the CuPt shell is coated on 206
Au, the CuPt (111) peak appears between Au (111) and CuPt 207
(111) peaks, indicating that the (111) lattice in the CuPt shell 208
is increased compared to that in pure CuPt NPs due to the Au 209
core effect. With more Cu present in the CuPt structure, the 210
(111) peak shifts from 39.5° (Au/Cu₃₅Pt₆₅), to 40.2° (Au/ 211
Cu₅₀Pt₅₀), to 40.8° (Au/Cu₆₄Pt₃₆), which follows Vegard's 212
Law¹² and proves the formation of CuPt alloy structure. 213
Incorporating more Cu in the CuPt shell led to a slight 214
increase in shell thickness. For example, Au/Cu₃₅Pt₆₅ has a 1 215
nm shell, and Au/Cu₆₄Pt₃₆ has a 1.5 nm shell. The CuPt 216
coating was further characterized by UV-vis spectroscopy and 217
electrochemical redox properties. The surface plasmonic 218
absorption peak of the 5 nm Au NPs in hexane dispersion 219
appeared at 520 nm. UV-vis spectra of the core/shell NP 220
dispersion in hexane showed no plasmonic absorption (Figure 221
3c). To test the electrochemical properties of these NPs, we 222
first deposited the NPs on the Ketjen carbon support by 223
sonicating for 1 h a mixture of NPs and carbon support in 5 mL 224
of hexane at a weight ratio of 1:2. We then used acetic acid and 225
0.1 M HClO₄ solution to wash the C-NP composite to remove 226
the surfactants around each NP.¹³ These combined acid 227
treatments did not change either Au/Pt ratio or Cu 228
composition. Cyclic voltammograms (CVs) of the NPs recorded 229
from 0.02 to 1.61 V in N₂-saturated 0.1 M HClO₄ is given in 230
Figure 3d. We can see that the C-Au NPs have a Au-O-related 231
reduction peak at 1.08 V, while C-Au/CuPt NPs show only 232
CuPt-O and no Au-O reduction peak, indicating that Au is 233
embedded under the CuPt shell. 234

Cu in the Au/CuPt NPs could be further etched away 235
electrochemically, forming Au/CuPt/Pt core/shell NPs. This 236
was similar to the electrochemical dealloying of Cu observed in 237
CuPt NPs.^{7a,14} Figure 4a lists several rounds of CV scans for 238 #

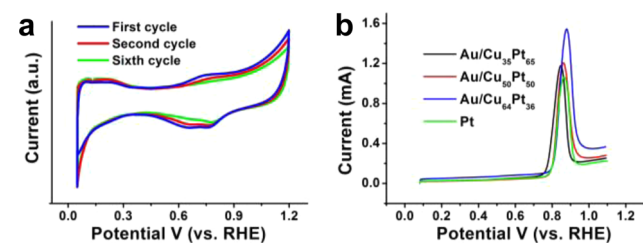


Figure 4. (a) CV curves of the C-Au/Cu₆₄Pt₃₆ catalyst during electrochemical dealloying at 100 mV/s. (b) CO stripping curves of the C-Au/Cu_xPt_{100-x} and C-Pt catalysts in 0.1 M HClO₄.

the Au/Cu₆₄Pt₃₆ NPs. In the first anodic scan, there is a broad 239
peak near 0.6–0.7 V which refers to the Cu dissolving process 240
from the CuPt shell. In the backward cathodic scan, there are 241
two peaks related to the reduction of Pt-O (0.8 V) and Cu-O 242
oxide (0.6 V). During the second scan, the area of the Cu 243
dissolving peak decreased dramatically and the hydrogen 244
desorption peak (0.04–0.4 V) increased, indicating the removal 245
of surface Cu and the formation of the Pt layer. Once the 246
dealloying reached equilibrium and there was no hydrogen 247
desorption peak increase (after 50 cycles), Au/Cu₅₀Pt₅₀ and 248
Au/Cu₆₄Pt₃₆ were converted to Au/Cu₃₆Pt₆₄ and Au/Cu₃₅Pt₆₅ 249
was converted to Au/Cu₂₉Pt₇₁. The CO stripping peak of the 250
dealloyed Au/Cu₃₆Pt₆₄ (Figure 4b) resembled that of the pure 251
Pt^{5a} and core/shell FePt/Cu/Pt nanorods,¹⁵ confirming that 252
Au/Cu₃₆Pt₆₄ NPs have a smooth Pt shell, forming Au/CuPt/Pt. 253

254 The electrocatalytic activity of these Au/CuPt (Au/CuPt/Pt)
255 NPs for the ORR was studied. Figure 5a shows the

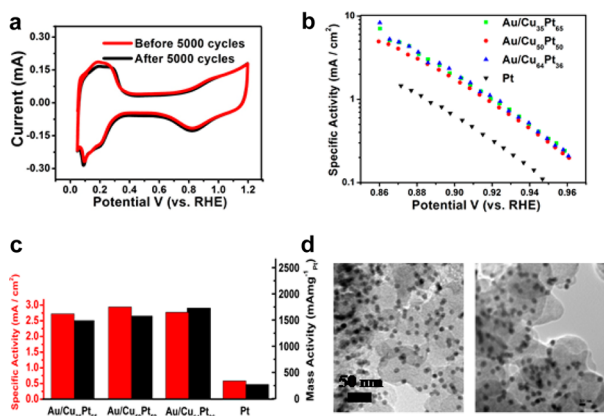


Figure 5. (a) CVs of C-Au/Cu₃₅Pt₆₅ NPs before and after 5000 potential cycles between 0.6 and 1.0 V. (b) Activity–potential plots of C-Au/Cu_xPt_{100-x} and C-Pt catalysts. (c) Specific and Pt mass activities of C-Au/Cu_xPt_{100-x} and C-Pt catalysts at 0.9 V. (d) TEM of C-Au/Cu₃₅Pt₆₅ NPs before (left) and after (right) 5000 potential cycles.

256 representative CV curves obtained in N₂-saturated 0.1 M
257 HClO₄ solution. The ORR polarization curves were obtained in
258 O₂-saturated 0.1 M HClO₄ solution. The Tafel plots of three C-
259 NPs (Au/Cu₃₅Pt₆₅, Au/Cu₃₀Pt₅₀, and Au/Cu₆₄Pt₃₆) are given in
260 Figure 5b. By comparing CV and ORR curves of different Au/
261 CuPt NPs, we noticed that the specific activities of the core/
262 shell NPs were not Cu composition dependent. This is likely
263 caused by the dealloying of Cu (leaving only 25–36% Cu
264 present in the final core/shell structure) and the formation of
265 the Pt shell. The specific activities of the Au/CuPt reach as high
266 as 2.75 mA/cm² at 0.9 V which is nearly five times higher than
267 that from the 5 nm commercial Pt (commercial TKK
268 TEC10E50E-HT 5 nm Pt 50.5% wt Pt loading) (0.58 mA/
269 cm²). Figure 5c lists the specific and mass activities of the core/
270 shell NPs and Pt NPs for ORR. The core/shell NPs have mass
271 activity up to 1700 mA/mg_{Pt} at 0.9 V (from the Au/Cu₆₄Pt₃₆
272 NPs), six times higher than that of the commercial 5 nm Pt
273 (270 mA/mg_{Pt}). The stability of the Au/CuPt NPs was also
274 tested by performing 5000 potential sweeps between 0.60 and
275 1.10 V at 100 mV/s in O₂-saturated 0.1 M HClO₄ solution.
276 Figure 5a shows the CVs of the Au/Cu₃₅Pt₆₅ NPs before and
277 after the test. The ECASA of the Au/CuPt dropped only 7.8%
278 after this stability test while that of the commercial Pt decreased
279 by 20%. TEM analysis further confirmed the stability of the Au/
280 CuPt NPs, as they showed little morphology change after the
281 stability test (Figure 5d).

282 The Au/CuPt NPs were also an active catalyst for MOR.
283 Figure 6 summarizes the CV oxidation and stability test results.
284 Among all core/shell NPs studied, the Au/Cu₆₄Pt₃₆ NPs
285 showed the highest catalytic activity (Figure 6a). Even after a 1
286 h stability test, the catalytic activity of the Au/Cu₆₄Pt₃₆ NPs was
287 still much higher than that of Pt (Figure 6b). This high activity
288 is attributed to the high CO tolerance of the Au/CuPt NPs as
289 characterized by the current ratio change between two peaks in
290 forward (*I_f*) and backward (*I_b*) scans in the CV; the larger the
291 ratio, the better the CO removal activity of the catalyst.¹⁶ The
292 mass activity of the Au/Cu₆₄Pt₃₆ NPs was calculated to be
293 around 441 mA/mg_{Pt}, four times higher than that of Pt NPs at
294 0.8 V.

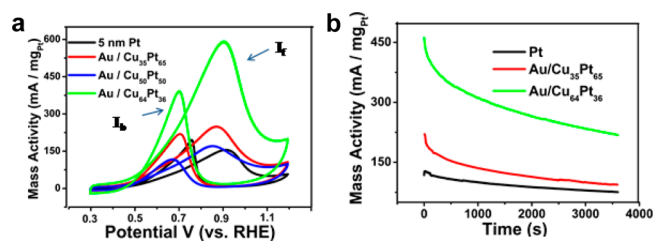


Figure 6. (a) MOR curves and (b) catalyst stability test at 0.8 V in 0.1 M HClO₄ + 0.1 M methanol.

The observed enhancement of the Au/CuPt NPs on both
295 ORR and MOR catalysis is likely caused by the combination of
296 alloy and strain effects present in the core/shell structure.
297 Similar to what has been predicted and observed, alloying Pt
298 with an early transition metal lowers the Pt d-band level,
299 weakening the binding of oxygenated spectator (blocking)
300 species (e.g., OH⁻) to Pt and increasing the number of active
301 Pt sites that are accessible to oxygen.¹⁷ The formation of Au/
302 CuPt/Pt may also have the surface Pt lattice compressed,
303 further favoring ORR.^{7a} The presence of electropositive Au
304 in the core can prevent Pt from being easily oxidized,^{5a,18}
305 stabilizing the Pt shell in the ORR condition. The Au core
306 and Cu alloying effects must also help methanol adsorption/
307 activation on Pt with high CO tolerance, rendering the Au/
308 CuPt/Pt NPs equally effective in catalyzing MOR.
309

In conclusion, we have reported a facile synthesis of
310 monodisperse 5/1.5 nm Au/CuPt NPs by coreduction of
311 Pt(acac)₂ and Cu(acac)₂ in a mixture of 5 nm seeding Au NPs,
312 HDD, OAm, OA, and ODE at 200 °C. The Au core provides
313 the proper nucleation sites for CuPt alloy formation, and as a
314 result, the nucleation/growth of the CuPt shell occurs at a
315 lower temperature without the formation of free CuPt NPs.
316 Due to the CuPt alloy effect and core/shell interactions, these
317 Au/CuPt NPs show promising dual catalysis for both ORR and
318 MOR in 0.1 M HClO₄ solution with their ORR specific (mass)
319 activities reaching 2.72 mA/cm² (1500 mA/mg_{Pt}) at 0.9 V and
320 MOR specific (mass) activities at 0.755 mA/cm² (441 mA/
321 mg_{Pt}) at 0.8 V. Our tests prove that the existence of the Au NP
322 core not only minimizes the Pt usage in its catalysis for fuel cell
323 reactions but also improves drastically the stability of the core/
324 shell catalyst. The results suggest that the core/shell design is
325 indeed effective for optimizing NP catalysis. The same concept
326 may be extended to other multimetallic NP system, making it
327 possible to tune NP catalysis for many different chemical
328 reactions.
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Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

This work was supported by the U.S. Army Research
336 Laboratory and the U.S. Army Research Office under the
337 Multi University Research Initiative (MURI, grant number
338 W911NF-11-1-0353) on “Stress-Controlled Catalysis via
339 Engineered Nanostructures”.
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